

# Calculation of phase diagrams for the CaO-CaF<sub>2</sub> and CaO-BaO binary systems by using molecular dynamics

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The thermodynamic properties for the CaO-CaF<sub>2</sub> and CaO-BaO systems were calculated by molecular dynamics (MD) simulation using Born-Mayer-Huggins type pair potential. The parameters of interatomic potential were obtained by fitting the thermodynamic properties of CaO, CaF<sub>2</sub> and BaO with the experimental data. The phase diagrams for the CaO-CaF<sub>2</sub> and CaO-BaO systems were calculated by the thermodynamic properties obtained by MD simulation. The eutectic composition and temperature of the CaO-CaF<sub>2</sub> system calculated by MD simulation were about 28 mole per cent CaO and 1575K, respectively. The calculated eutectic point agrees very well with experimental data, which are about 22 mole per cent CaO and 1633K, respectively.

Keywords: molecular dynamics simulation, thermodynamics, phase diagram, calcium oxide, calcium fluoride, barium oxide.

## Introduction

Recently, the thermodynamic properties and phase diagrams of multiphase molten oxides are generally obtained by thermodynamic calculation such as FactSage, Thermo-Calc and ChemSage software. However, the application of these calculation methods is very limited because the experimentally obtained thermodynamic data are required for the calculation of thermodynamic properties of multiphase molten oxides. Molecular dynamics (MD) simulation can be used to calculate the thermodynamic properties based on the dynamic quantities of individual particles of solids and fluids without the thermodynamic database. Therefore, the thermodynamic properties of various systems can be calculated for the system, where experiments are difficult.

The CaO-based slag system such as the CaO-CaF<sub>2</sub> and CaO-BaO-CaF<sub>2</sub> systems are generally used in steelmaking process. There have been many investigations about the physical and chemical properties. Barium oxide-based slags have been attractive with the possibility of application in hot metal pretreatment making use of their high basicity and low melting temperature. However, in spite of importance of these slag systems, thermodynamic properties and phase diagram of barium-based systems have remained obscure because of the difficulties associated with experimental measurements.

The purpose of the present research is to calculate the thermodynamic properties and phase diagram of the CaO-CaF<sub>2</sub> and CaO-BaO binary systems using MD simulation with optimum parameters obtained by fitting the thermodynamic properties of CaO, CaF<sub>2</sub> and BaO.

## Molecular dynamics calculation

### Interatomic potential

The potential energy was calculated by the sum of pairwise interactions between atoms in the Born-Mayer-Huggins type pair potential according to Equation [1].

$$\phi_{ij} = Z_i \cdot Z_j e^2 / r_{ij} + f_0 (b_i + b_j) \exp\left\{(\sigma_i + \sigma_j - r_{ij}) / (b_i + b_j)\right\} \quad [1]$$

where  $r_{ij}$  is the interatomic distance between ions  $i$  and  $j$ ,  $Z_i$  is the valence of the ion  $i$ ,  $e$  is the electron charge,  $f$  is a standard force of  $4.184 \text{ kJ} \text{ \AA}^{-1} \text{ mol}^{-1}$ ,  $\sigma_i$  and  $b_i$  are repulsive radius and softness parameters of the ion  $i$ . The interatomic pair potential terms represent Coulombic and repulsion interactions without the dispersion terms.

Belashchenko *et al.*<sup>1-2</sup> and Hirao *et al.*<sup>3</sup> calculated the interatomic potential parameters for the CaF<sub>2</sub>, CaO and BaO systems. These interatomic potential parameters show good agreement with the structural properties obtained by the experimental methods. However, these parameters have a limitation for the calculation of thermodynamic properties such as melting point and fusion data of the CaF<sub>2</sub>, CaO and BaO system.

In the present study, interatomic potential parameters for Ca and O were taken from Matsumiya *et al.*<sup>4</sup> which reproduces the various properties such as thermodynamic, structural and transport property of CaO solid and CaO-SiO<sub>2</sub> melt very well. The optimum parameters for Ba and F were calculated by fixing the interaction parameters of Ca and O. The interatomic potential parameters used in this study are listed in Table I.

### Methods for calculation

The MD simulation was carried out using the isobaric and isothermal ensemble. Temperature is controlled by velocity scaling method. Pressure is controlled by Parrinello and Rahman method at atmospheric pressure.

The CaO and BaO crystals have a cubic rocksalt structure. The CaF<sub>2</sub> crystal has a CaF<sub>2</sub> structure. The MD basic cell for CaO, BaO and CaF<sub>2</sub> solid phase is composed with 5×5×5 unit cell of CaO, BaO and CaF<sub>2</sub> crystals in the cubic cell. The total number of atoms in CaO, CaF<sub>2</sub> and BaO were 1000, 1500 and 1000. In the case of the liquid phase of CaO, CaF<sub>2</sub> and BaO, the initial atomic

configurations were set to be random in the cubic cell. The total number of atoms in CaO, CaF<sub>2</sub> and BaO were 1000, 1500 and 1000. The initial atomic configurations for the CaO-CaF<sub>2</sub> and CaO-BaO binary systems were set to be random in the cubic cell. The total number of atoms was taken to be from 1000 to 1500 with changing composition of the binary systems. The densities of basic cells for the CaO-CaF<sub>2</sub> and CaO-BaO systems of 3.18–3.34 g/cm<sup>3</sup> and 3.34–5.72 g/cm<sup>3</sup> were used. The periodic boundary condition was employed for each simulation. The long range Coulomb interactions have been summated by Ewald method. The equations of motion were integrated with fifth-order Gear's predictor-corrector algorithms with a time step of 1fs. Each MD run durations involved 10 000 steps. The thermodynamic properties were calculated by statistical analyses of velocities and position data after reaching equilibrium for the system at the various temperatures. The MD simulation was carried out using WinMASPHYC program.

## Results and discussion

### CaO, CaF<sub>2</sub> and BaO system

The enthalpies of solid and liquid phase for CaO, CaF<sub>2</sub> and BaO were calculated as a function of temperature at the atmospheric pressure. Figure 1 shows the calculated and observed<sup>5</sup> enthalpy for CaO, CaF<sub>2</sub> and BaO as a function of temperature. The values of the enthalpy are calculated for a reference temperature of 1000K by neglecting quantum corrections in this study. Table II shows the melting point, enthalpy and entropy of fusion of CaO, CaF<sub>2</sub> and BaO calculated by MD simulation compared with the observed data<sup>5</sup>. The thermodynamics properties of CaO, CaF<sub>2</sub> and BaO calculated by the MD simulation are, on the whole, in good agreement with experimentally observed data. The potential equation and parameters used in the present study are reasonable for the calculation of thermodynamic properties of the CaO, CaF<sub>2</sub> and BaO system.

Table I

Parameters of interatomic potential used for the simulation

	Z <sub>i</sub>	σ <sub>i</sub> (°)	b <sub>i</sub> (°)
Ca	2	1.9995	0.2101
Ba	2	2.55	0.2685
O	-2	1.84	0.13
F	-1	1.4848	0.116

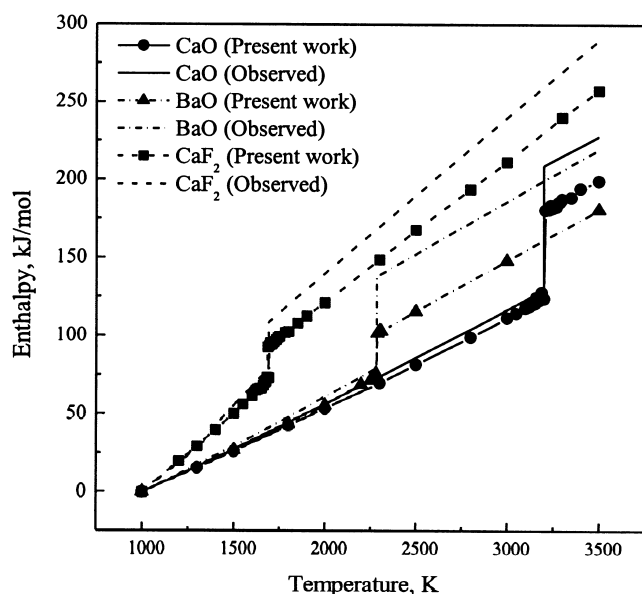


Figure 1. Variation of calculated and observed enthalpy of CaO, CaF<sub>2</sub> and BaO at atmospheric pressure as a function of temperature. (Reference state: 1000K)

### The CaO, CaF<sub>2</sub> and CaO-BaO binary systems

#### Calculation of Gibbs energy of mixing

The enthalpy of mixing of CaO-CaF<sub>2</sub> and CaO-BaO binary systems directly calculated by MD simulation at various temperatures and compositions. The enthalpy of mixing is expressed by Equation [2].

$$\Delta H^M = (X_A \bar{H}_A + X_B \bar{H}_B) - (X_A H_A^o + X_B H_B^o) \quad [2]$$

where  $\bar{H}_i$  and  $H_i^o$  are the partial molar enthalpy of component  $i$  in the solution and the standard molar enthalpy of component  $i$ , respectively. Figure 2 shows the enthalpy of mixing curves of the CaO-CaF<sub>2</sub> binary system calculated by MD simulation at various temperatures. The enthalpy of mixing curve of the CaO-CaF<sub>2</sub> system shows the negative values in a composition and it does not show temperature dependence.

The internal energy and enthalpy of the systems can be calculated by the MD simulation as a function of particle positions and velocities. However, the entropy cannot directly be calculated by the MD simulation. Therefore, in this study, assuming that CaO-CaF<sub>2</sub> and CaO-BaO binary melts are complete ionic solutions and all the ions of CaO-CaF<sub>2</sub> and CaO-BaO binary melts have a random configuration, the entropy was calculated. The entropy of mixing can be calculated from ion fractions of solution by Equation [3].

$$\Delta S^M = -R(X_i \ln X_i + X_j \ln X_j + X_k \ln X_k) \quad [3]$$

where the  $X_i$ ,  $X_j$  and  $X_k$  are fractions of ions in the solution. Figure 3 shows the entropy of mixing of CaO-CaF<sub>2</sub> and CaO-BaO binary systems.

The Gibbs energy of mixing of Equation [4] is calculated

Table II

Comparison of observed and calculated thermodynamics properties of CaO, CaF<sub>2</sub> and BaO

	CaO		CaF <sub>2</sub>		BaO	
	Observed	Present work	Observed	Present work	Observed	Present work
Melting point (K)	3200±50	3210±10	1691±5	1695±5	2285±15	2285±5
$\Delta_{fus}H^o$ (kJ mol <sup>-1</sup> )	79.5	55.1	29.7	22.25	58.6	28.3
$\Delta_{fus}S^o$ (J K <sup>-1</sup> mol <sup>-1</sup> )	24.8	17.2	17.6	13.1	25.6	12.4

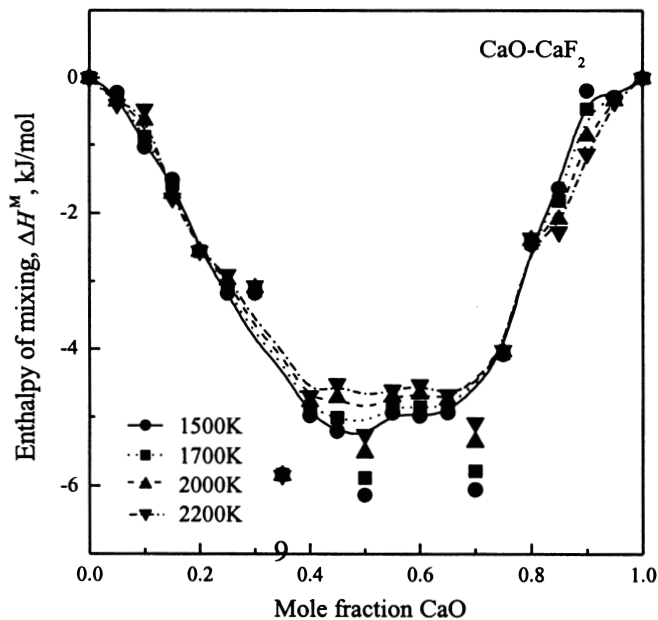


Figure 2. Enthalpy of mixing curves calculated at various temperatures for the CaO-CaF<sub>2</sub> binary system as a function of composition

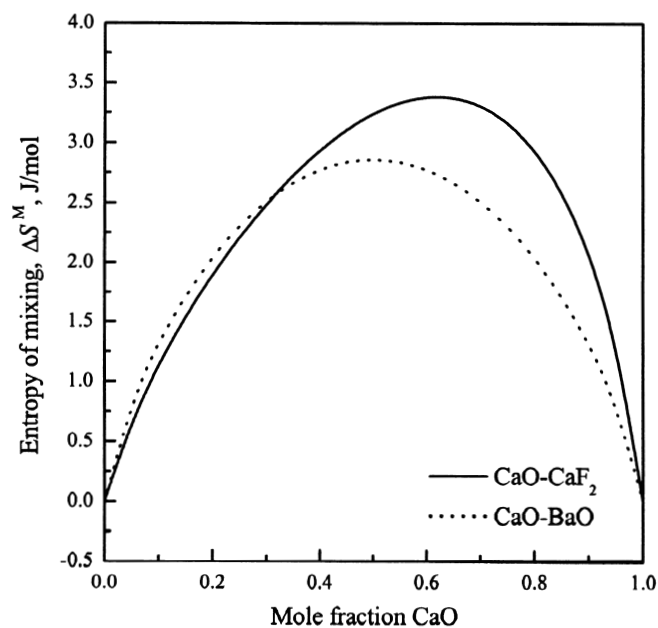


Figure 3. Calculated entropy of mixing curves of CaO-CaF<sub>2</sub> and CaO-BaO binary systems as a function of composition

by the values of enthalpy of mixing obtained by the MD simulation and the entropy of mixing calculated by Equation [3].

$$\Delta G^M = \Delta H^M - T\Delta S^M \quad [4]$$

Figure 4 shows the Gibbs energy of mixing of the CaO-CaF<sub>2</sub> system as a function of composition at various temperatures.

#### Calculation of the phase diagram for the CaO-CaF<sub>2</sub> system

The phase diagrams for the CaO-CaF<sub>2</sub> system is calculated by double tangent method of Gibbs energy of mixing.

Figure 5 shows the phase diagram of CaO-CaF<sub>2</sub> system calculated by MD simulation compared with the observed results<sup>6</sup>. The calculated eutectic composition and temperature for the CaO-CaF<sub>2</sub> system are 28 mole per cent CaO and 1573K, respectively. The eutectic point calculated by MD simulation was in good agreement with observed result of 22 mole per cent CaO at 1633K.

#### Conclusions

The optimum interatomic potential parameters of Ca, Ba, O and F were calculated for the MD simulation of the CaO-CaF<sub>2</sub> and CaO-BaO systems. The thermodynamics properties, such as melting point and enthalpy for solid and

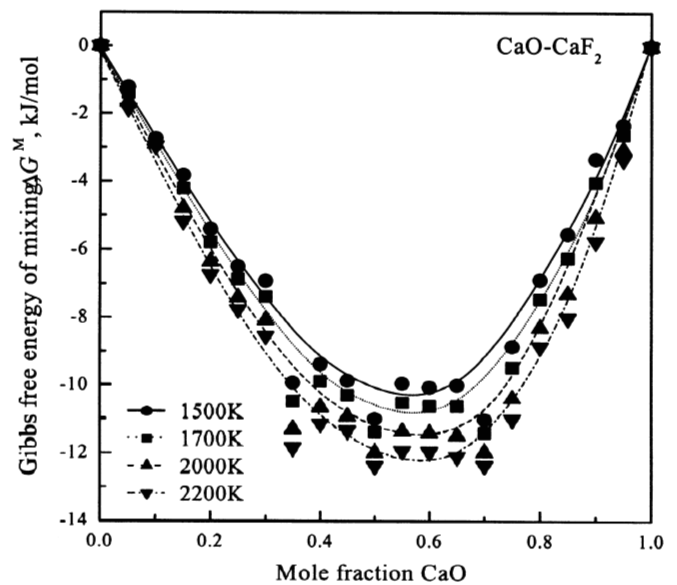


Figure 4. The calculated Gibbs energy of mixing at various temperatures for the CaO-CaF<sub>2</sub> binary system as a function of composition

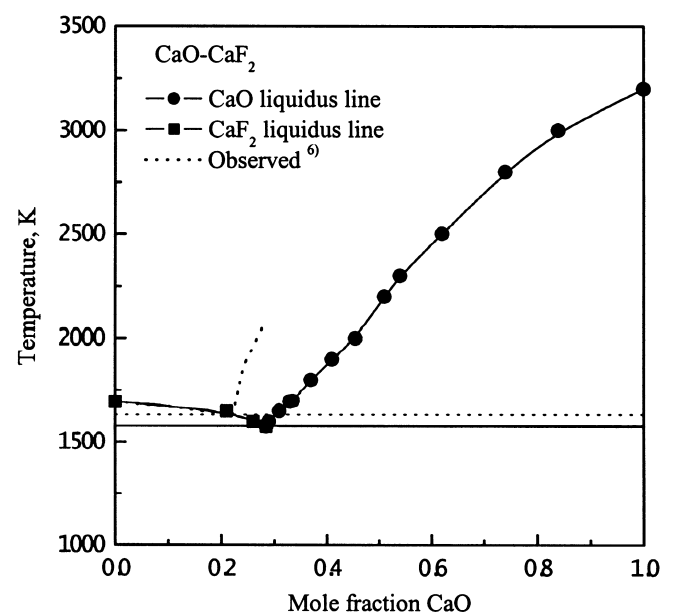


Figure 5. Calculated phase diagram of CaO-CaF<sub>2</sub> binary system with observed result

liquid phases of the CaO, CaF<sub>2</sub> and BaO, calculated by using potential parameters were in good agreement with observed values. The phase diagrams for the CaO-CaF<sub>2</sub> system were calculated by double tangent method of Gibbs energy of mixing. The eutectic composition and temperature for the CaO-CaF<sub>2</sub> system were calculated by MD simulation to be 28 mole per cent CaO and 1573K, respectively. The calculated eutectic point was in good agreement with observed result. It is concluded that the MD method can be used for the calculation of thermodynamic data for molten oxide systems and the estimation of phase diagrams of oxides at high temperature.

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